



# Damage prediction in rail welds

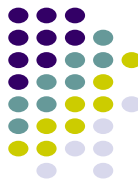
*Peter Mutton*

*ICRI-RCF Workshop, February 1, 2017*



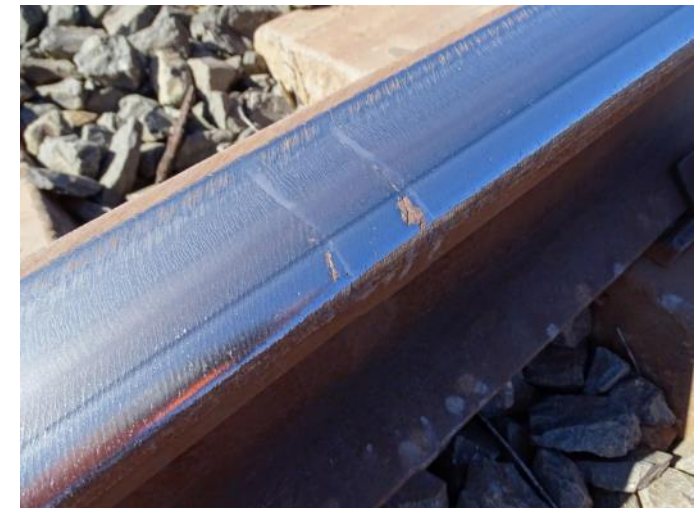
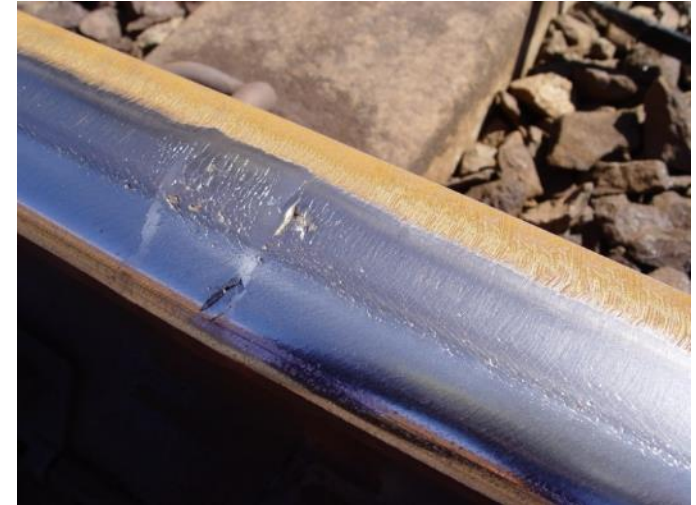
# Outline

- Rolling contact fatigue damage at welds
  - What is it
  - Why is a problem
- Initiation and growth of RCF damage
- Options for addressing the RCF damage
- Research questions and proposed approach



# Background

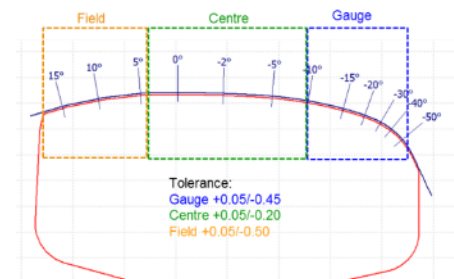
- Localised gauge corner damage in flashbutt welds in premium rail grades
  - More severe RCF damage in the heat-affected or softened zones
- Sensitivity to damage appears to vary between rail grades
- Observed under heavy haul conditions
  - Axle loads 35-40 tonnes
  - High adhesion locomotives





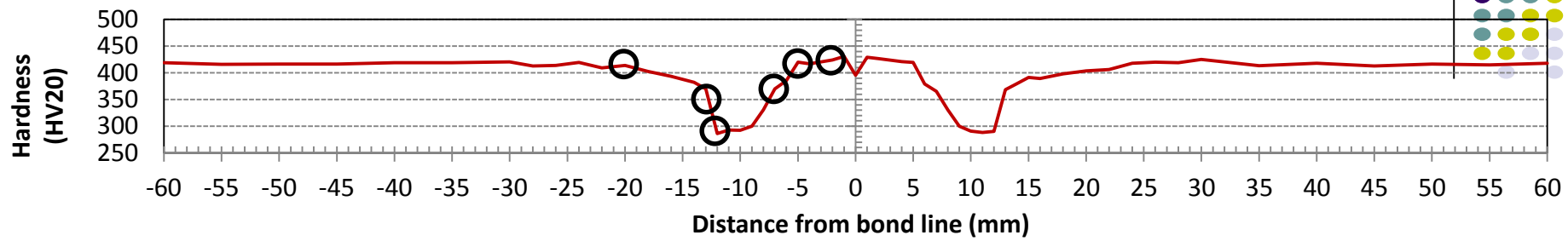
# Why is it an issue?

- Potential for development of transverse defects, particularly in high head loss rails
  - May restrict rail wear limits
- Additional or modified rail grinding procedures required to minimise or limit the extent of damage
- Limits the potential advantages of using premium rail grades

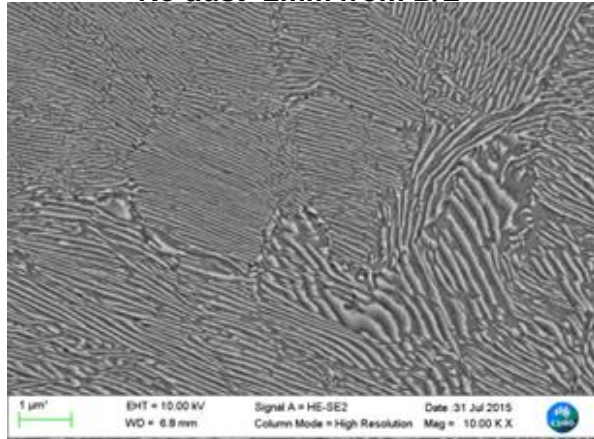




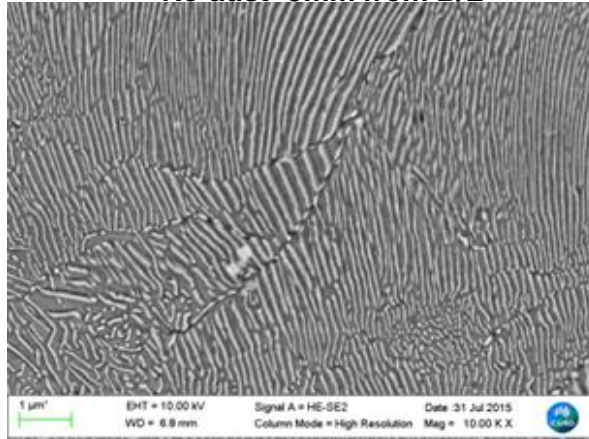
# Hardness/microstructure distribution



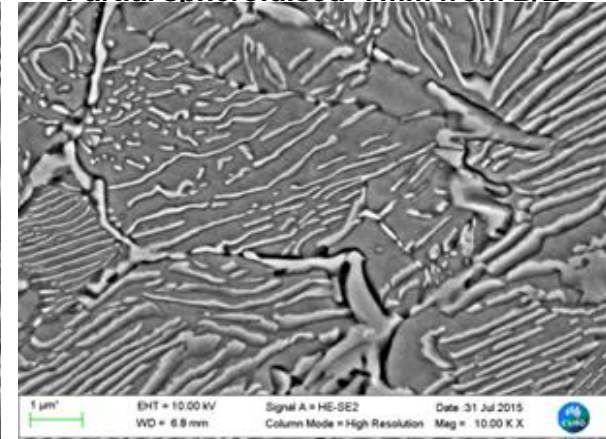
Re-aust -2mm from B/L



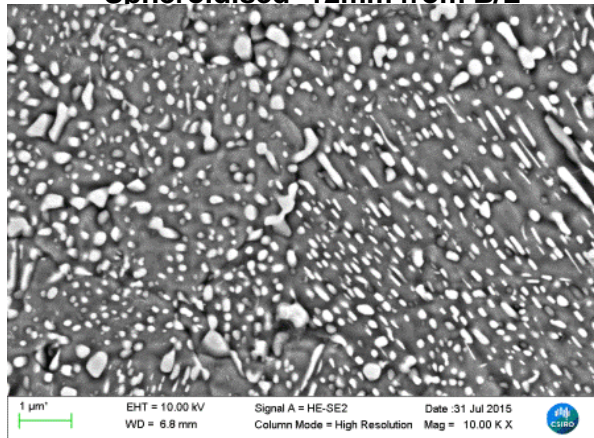
Re-aust -5mm from B/L



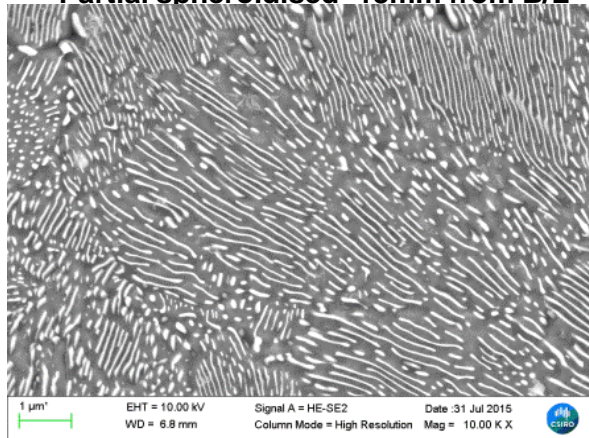
Partial spheroidised -7mm from B/L



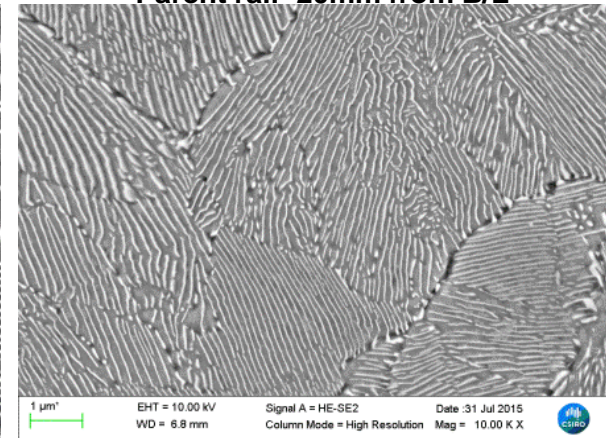
Spheroidised -12mm from B/L



Partial spheroidised -13mm from B/L



Parent rail -20mm from B/L

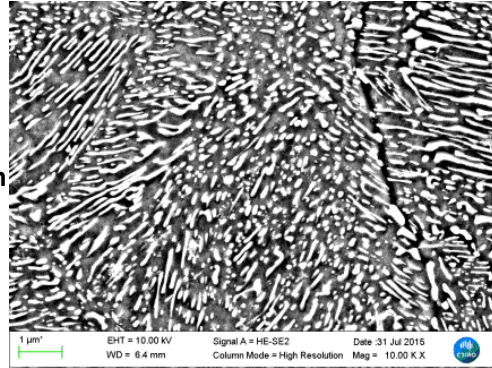




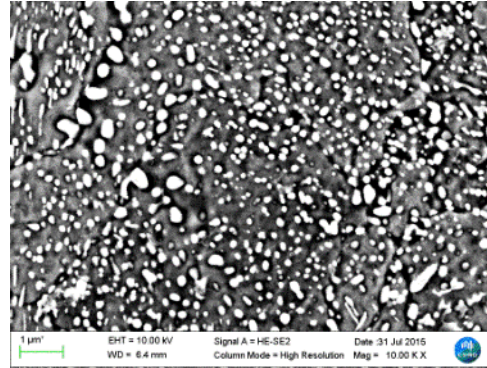
# New welds



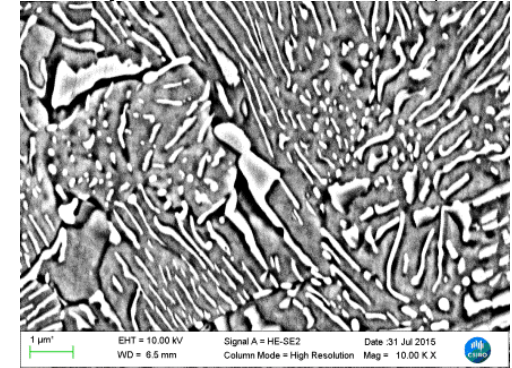
Partial spheroidised towards parent rail



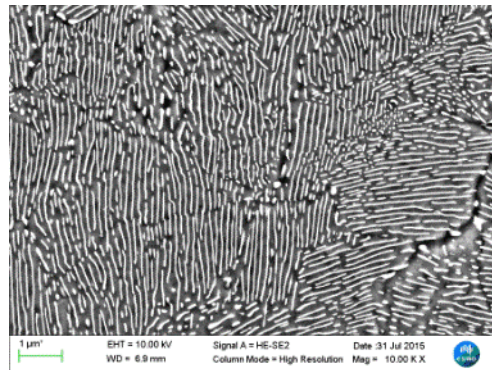
Spheroidised -12mm from B/L



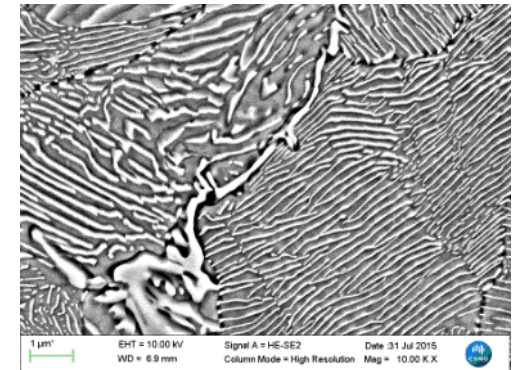
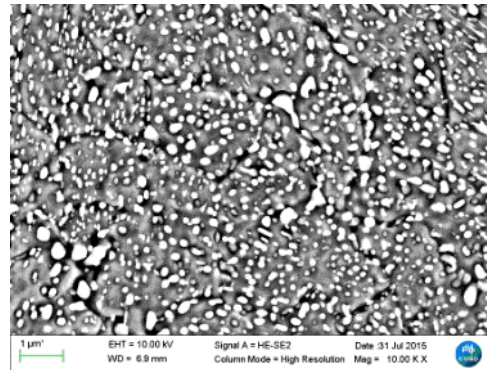
Partial spheroidised towards bond line



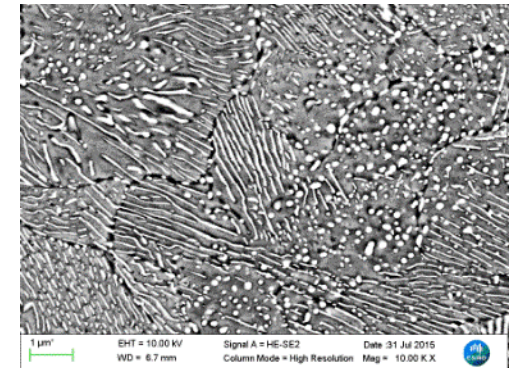
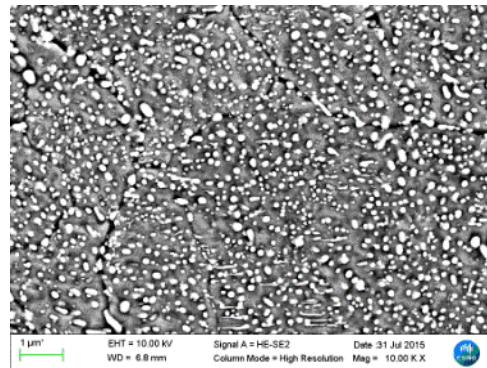
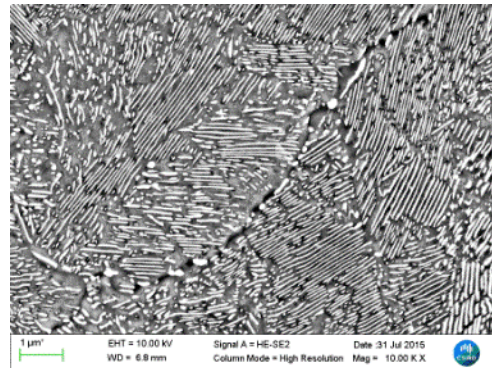
HEx Type C  
(mobile) 10mm  
below RS



HEx Type A  
(fixed) 20mm  
below RS



SP3 Type A  
(fixed) 20mm  
below RS

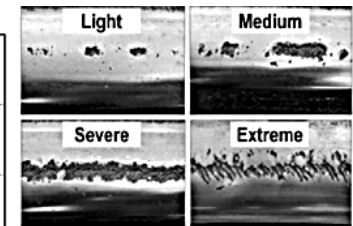
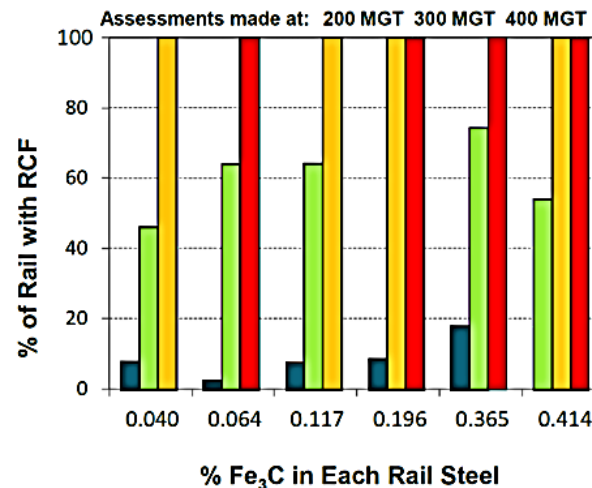
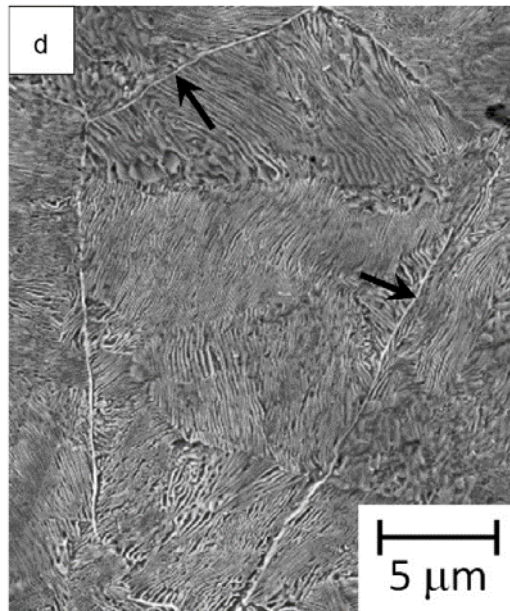






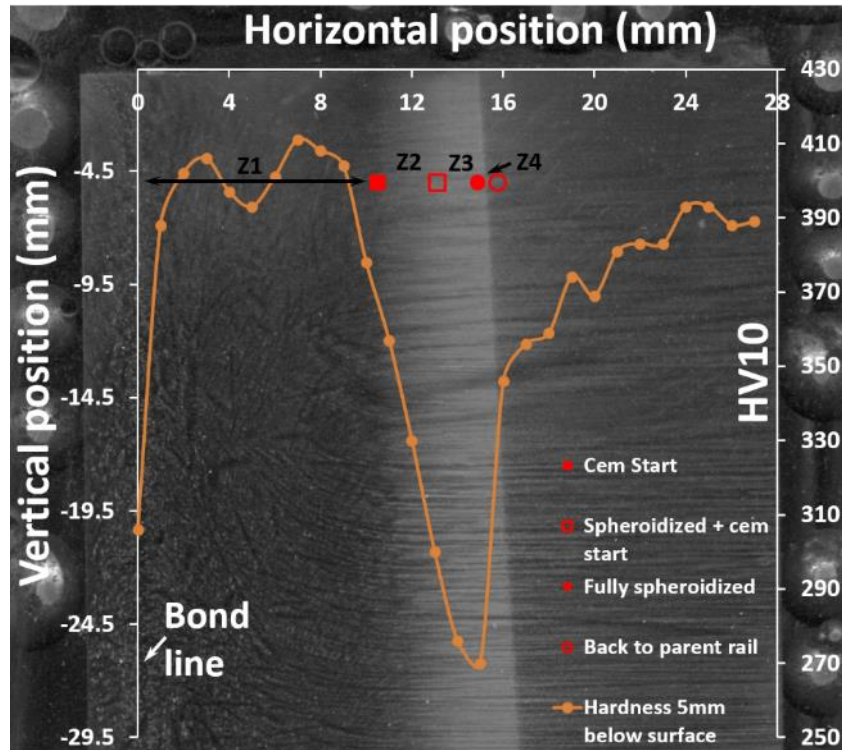
# Pro-eutectoid cementite

- A further consideration is the potential development of pro-eutectoid cementite ( $\text{Fe}_3\text{C}$ ) networks in welds manufactured in hypereutectoid premium rails
- The existence of pro-eutectoid  $\text{Fe}_3\text{C}$  at prior austenite grain boundaries has detrimental effects on fracture toughness and ductility, and may be linked to the development of RCF in rail steels [1]



[1] Gutscher D, Baillargeon J, Li D: *Railway Track & Structures*, 2014, vol.110, pp. 11-13..

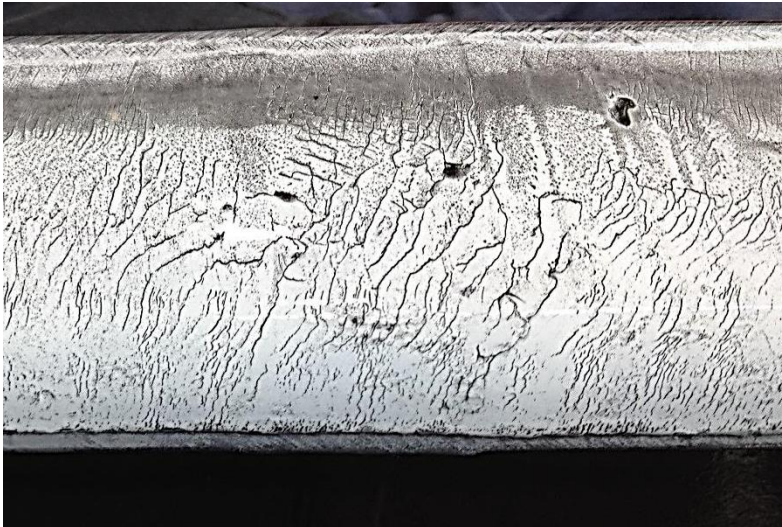
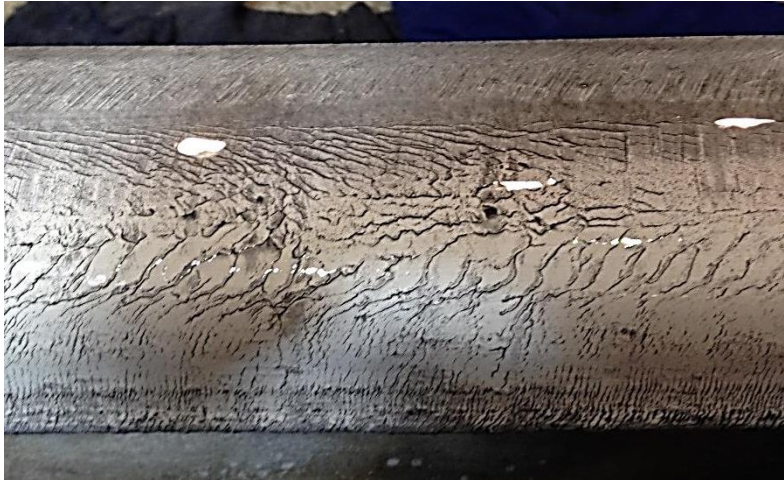
# New welds



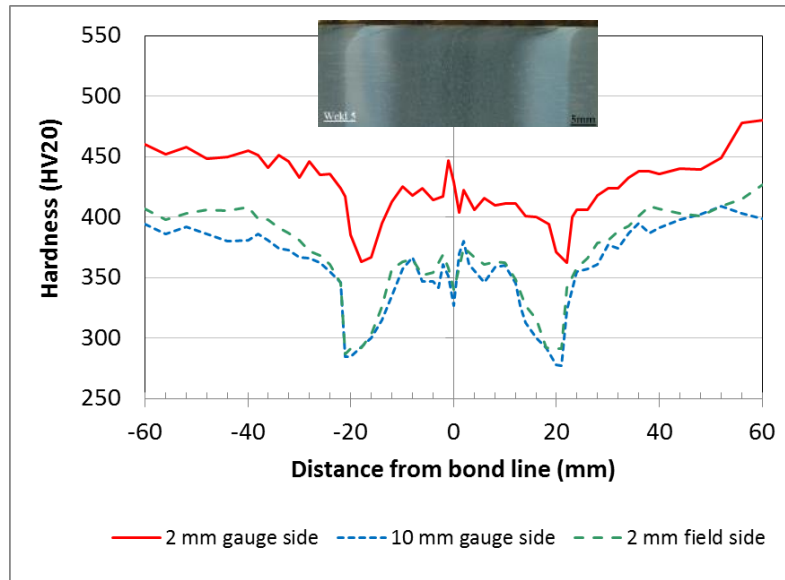
- Zone 1: no significant amount of grain boundary cementite observed.
- Zone 2: grain boundaries are heavily occupied by cementite
- Zone 3: grain boundary cementite and spheroidised microstructure are co-existing.
- Zone 4: fully spheroidised region



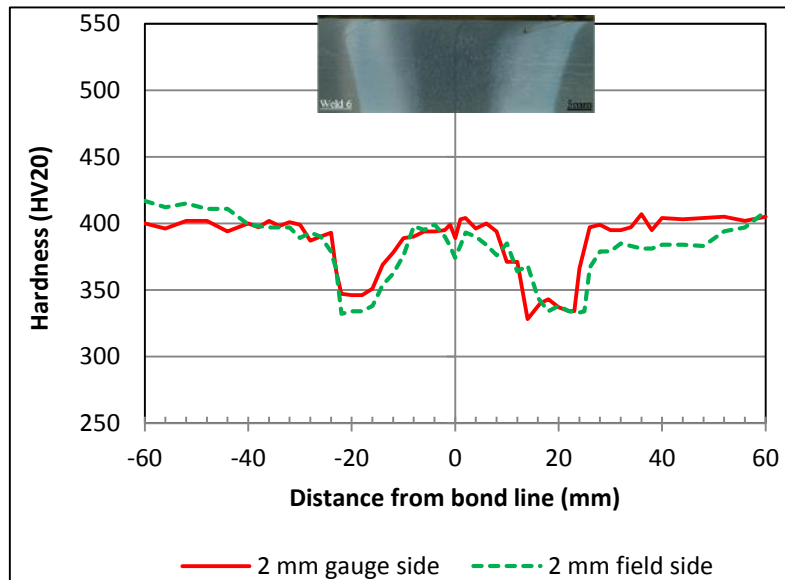
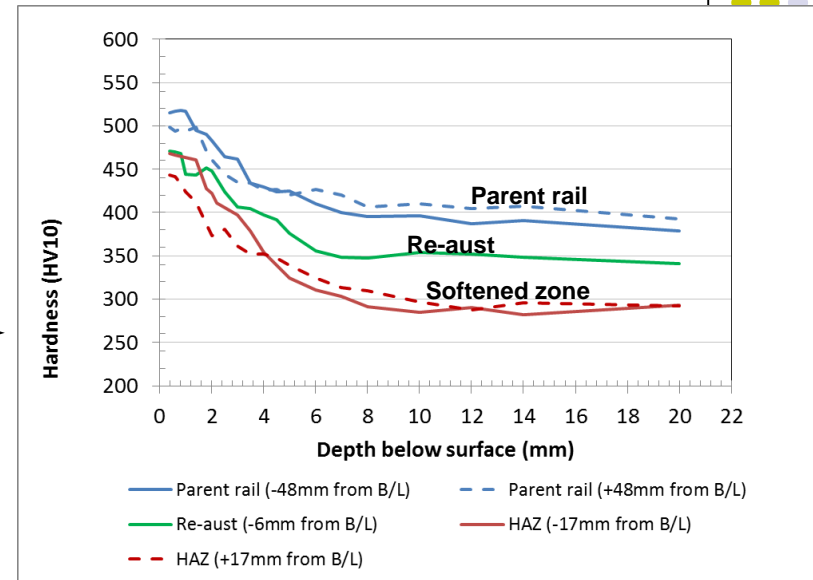
# Damage appearance: Ex-service welds



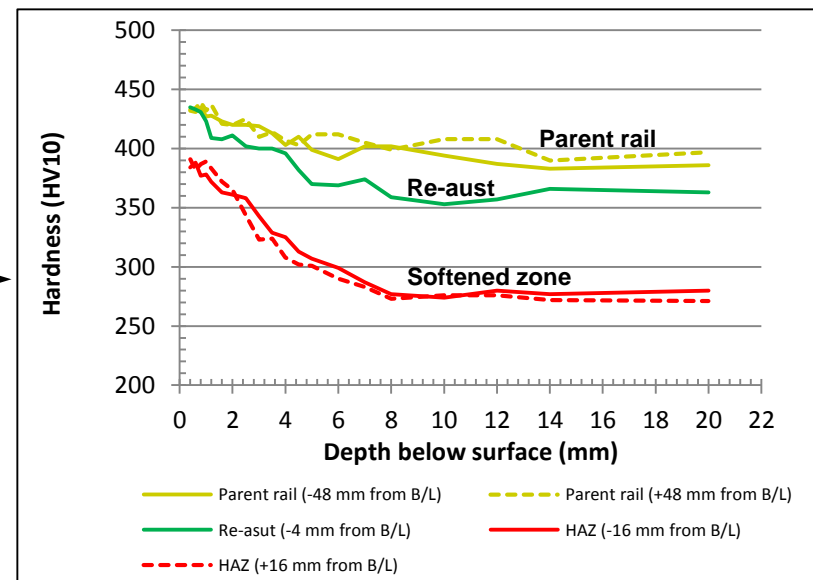
# Hardness: Ex-service welds



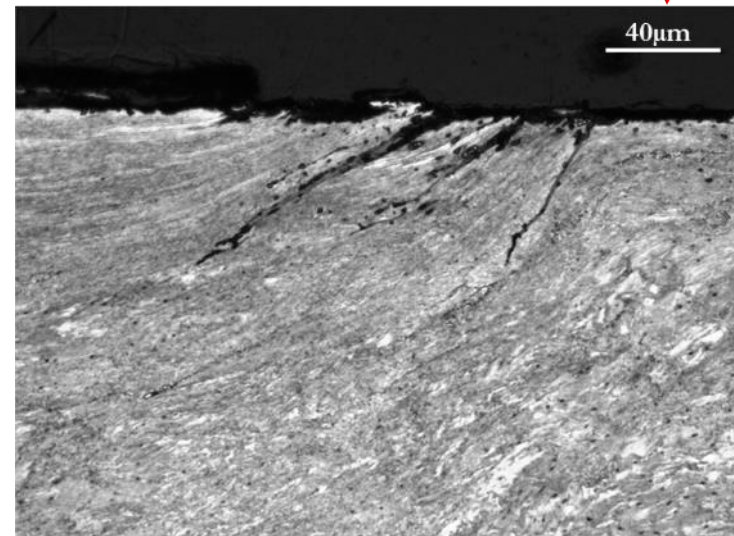
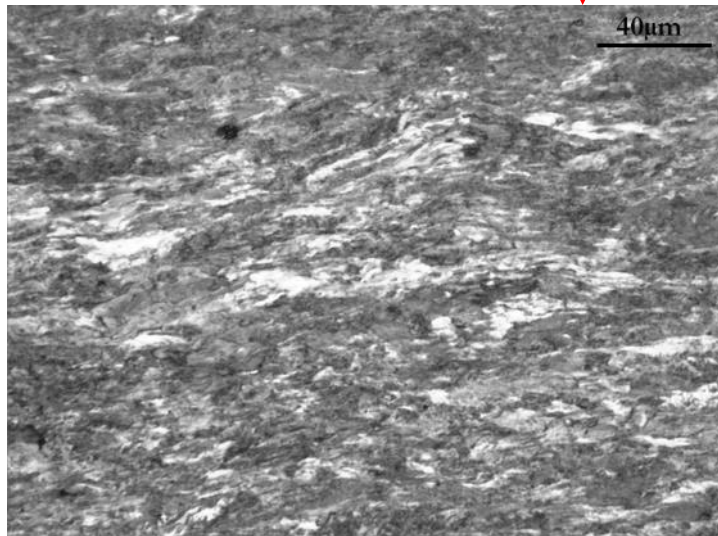
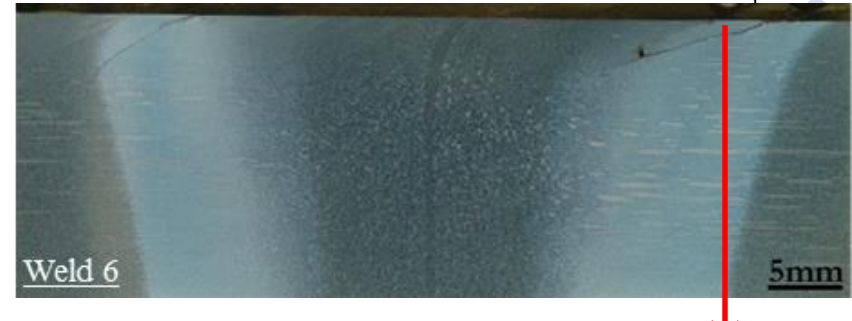
Weld 5



Weld 6



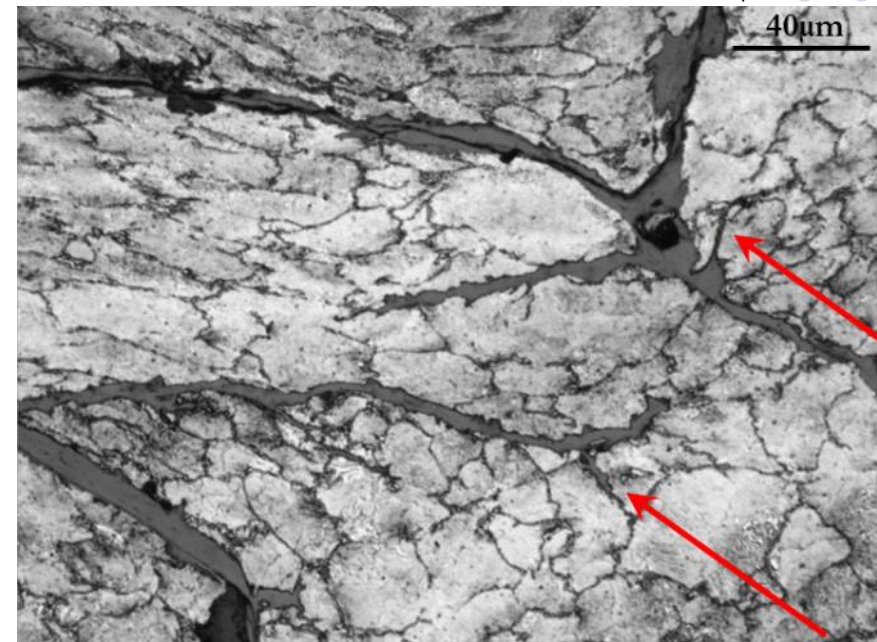
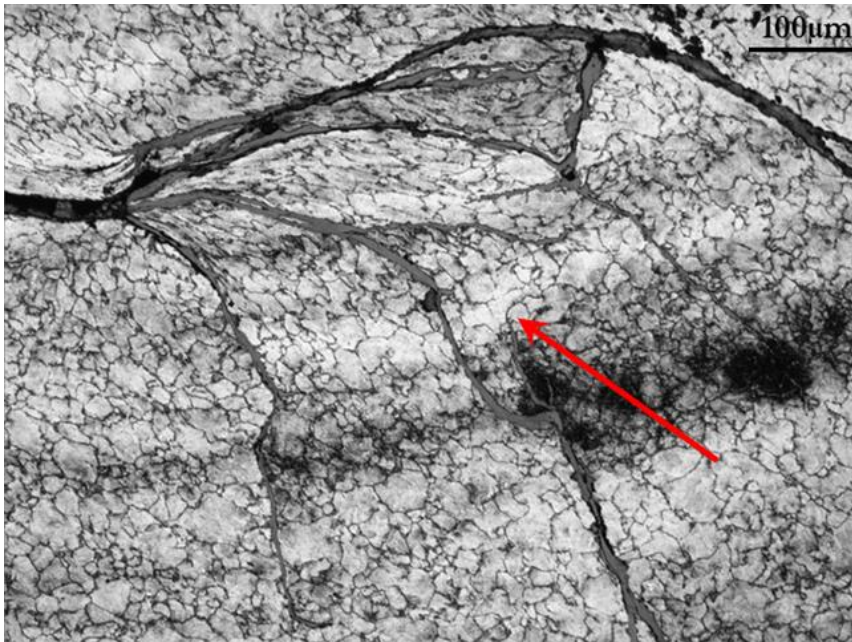
# Ex-service welds: Crack initiation



- Heavily deformed microstructure near the surface.
- Cracks penetrated deeper in the softened zones



# Ex-service welds: Crack growth



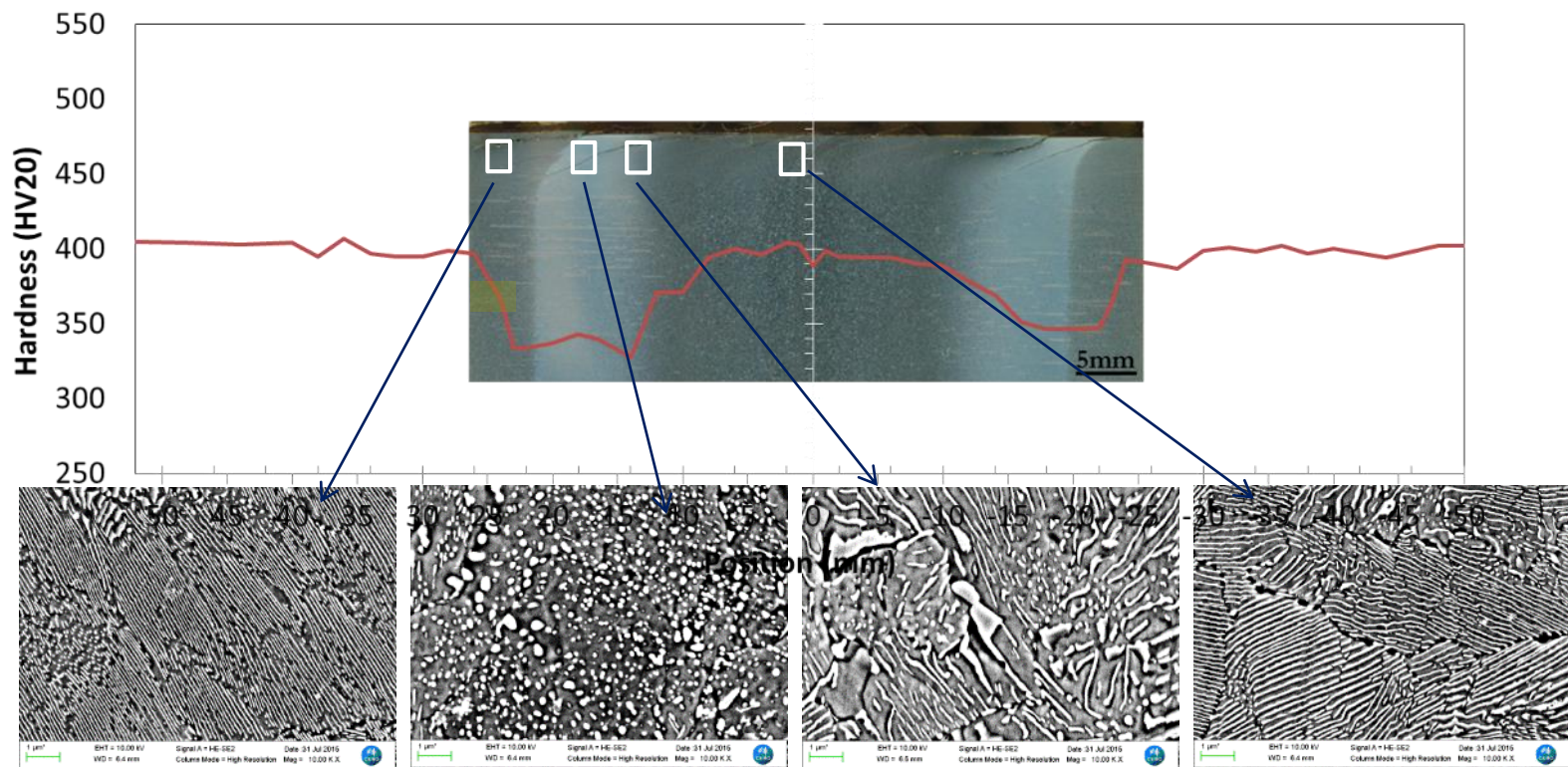
Weld 6

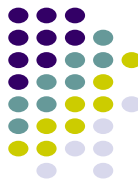
- Grain boundary  $\text{Fe}_3\text{C}$  does has a minor effect on crack propagation
  - Some crack tips penetrated along grain boundary, but most crack paths transgranular.



# Initiation of gauge corner damage

- Cracking initiates in the lower hardness material in the softened zone of the welds
  - Increased sensitivity cracking due to lower hardness or yield strength *and* limited work-hardening capacity of the spheroidised microstructure





# Propagation of cracks

- Cracks propagate more readily in softened zone due to greater depth of plastic deformation
- Direction of crack propagation can change at ~5mm below the surface, resulting on a transverse defect growth mode
- More extensive damage (spalling) develops on down side of welds (in direction of loaded train travel)

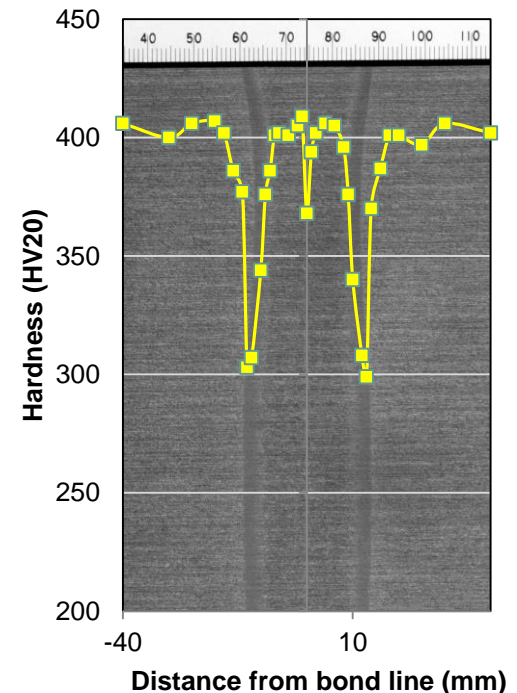






# Options for addressing the problem

- Modify the welding process to make HAZ less prone to gauge corner cracking
- Use rail grades which are more resistant to softening during welding, and have a lower tendency to develop pro-eutectoid cementite.
- Modify wheel-rail contact conditions to lower contact stress and creepage levels





# Research questions

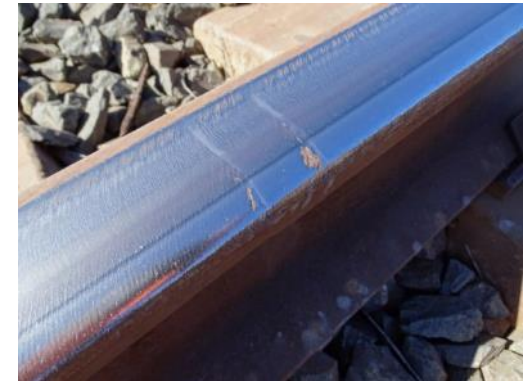
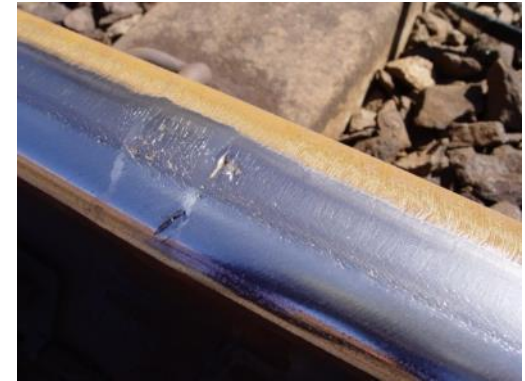
- What is the influence of the rail grade and welding conditions on the extent of variation in microstructure and mechanical properties between the softened zone and the parent rail?
- Do we have adequate material damage models for the range of microstructures that are present in rail welds?

# Plastic deformation and localised surface damage of rail flashbutt welds in heavy haul railway systems



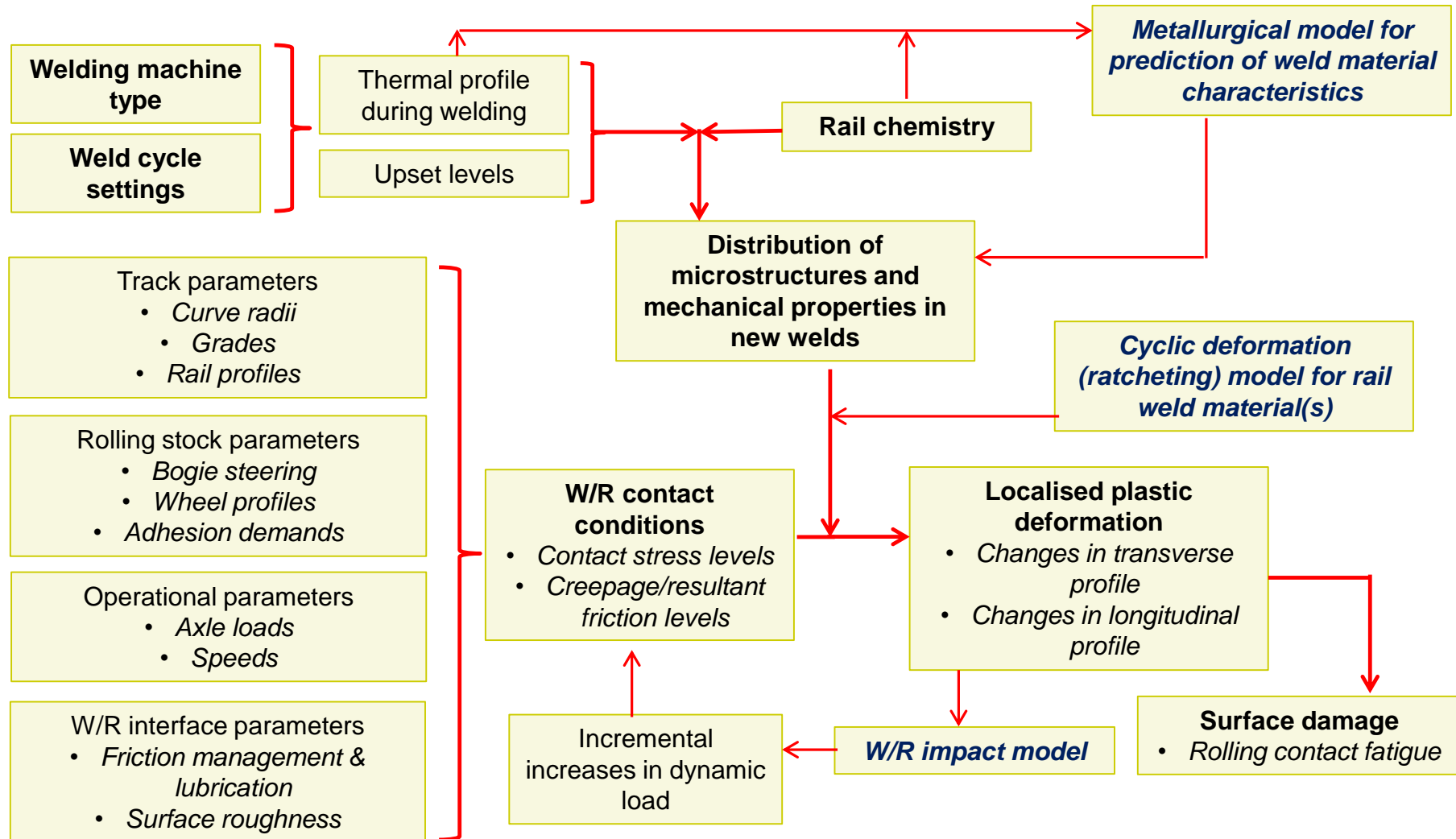
## Overall aim:

- To reduce the extent of localised surface damage at rail flashbutt welds in premium rail steels under heavy haul conditions through a combination of:
  - Developing a reliable tool to predict the extent of localised damage at rail welds, taking into consideration all of the influencing factors;
  - Optimising the combination of steel chemistry/grade and welding conditions to produce welds with an improved distribution of microstructures and mechanical properties;
  - Modifications to wheel/rail contact conditions and the associated rail maintenance procedures taking into consideration the cyclic deformation behaviour of rail welds.





# Plastic deformation and localised surface damage of rail welds in heavy haul railway systems





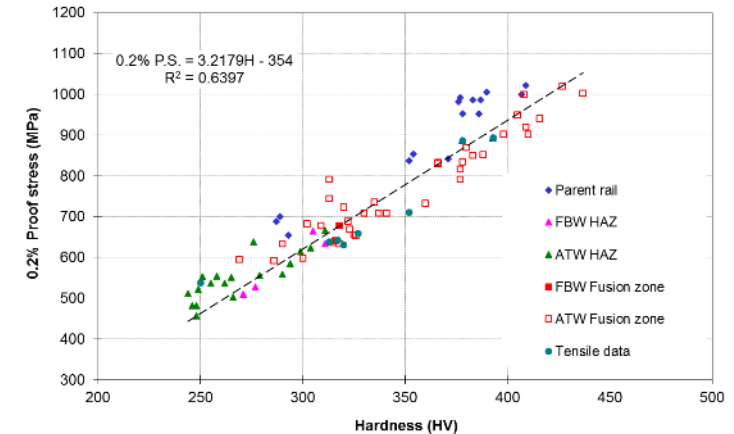
# Key tasks

- ***Metallurgical model for prediction of weld material characteristics***
  - *Characterisation of rail welds*  
Measurement and analysis of mechanical properties in new flashbutt welds at macroscopic and microscopic scale
  - *Characterisation of rail welding*
    - Determine the thermo-mechanical processing conditions involved in flashbutt welding
- ***Cyclic deformation (ratcheting) model for rail weld material(s)***
  - *Characterisation of deformation and damage at rail welds*
    - Measurement and characterisation of surface and subsurface deformation and damage in rail flashbutt welds which have been in service under heavy haul conditions
  - *Experimentally study the ratcheting behaviour (accumulation of plastic deformation) of rail welds under cyclic loading conditions*
    - Develop a methodology for reproducing the range of microstructures/mechanical properties that are present in flashbutt welds; use these to produce a representative range of test materials
    - Monotonic and cyclic deformation testing to develop ratcheting parameters for rail welds
  - *Develop a model that predicts cumulative plastic strain in rail welds*
    - Develop a numerical analysis methodology that incorporates the distribution of mechanical properties in rail welds, the associated ratcheting parameters, wheel-rail contact conditions and dynamic load effects to develop a multi-axial plastic deformation model for rail welds.

# Current status



- Previous compression tests to establish correlation between hardness and monotonic yield strength for parent rail and weld regions [1]
- Initial project to develop a preliminary approach for prediction of cyclic deformation parameters from monotonic test data
- New PhD project commencing March 2017



1. Mutton P, Cookson J, Qiu C, Welsby D (2015), *Wear* (2016), [Volumes 366–367](#), Pages 368–377.



# Preliminary estimation of cyclic deformation behaviour



- Microstructure  $\Leftrightarrow$  cyclic deformation behaviour
- Monotonic strength and hardness  $\Leftrightarrow$  cyclic deformation behaviour
- Estimate:
  - Cyclic deformation parameters
    - Cyclic strength coefficient  $K'$
    - Cyclic strain hardening exponent  $n'$
    - Cyclic yield strength  $S_y'$
  - From
    - Hardness (HB)
    - Yield strength  $S_y$
    - Ultimate tensile strength  $S_u$
    - Elastic modulus  $E$



# Questions?