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ICRI Presentation: *Defect Growth Characterization in Modern Rails*

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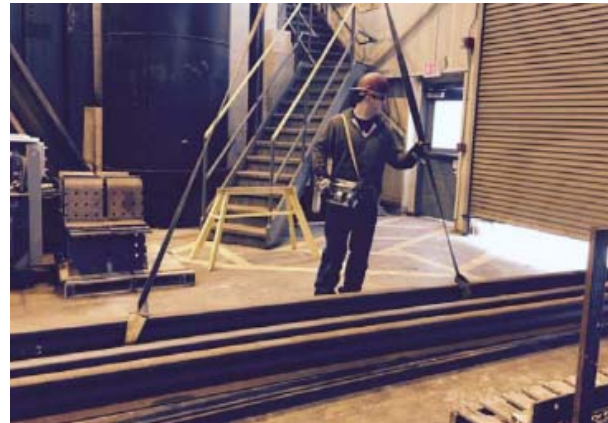


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INTRODUCTION

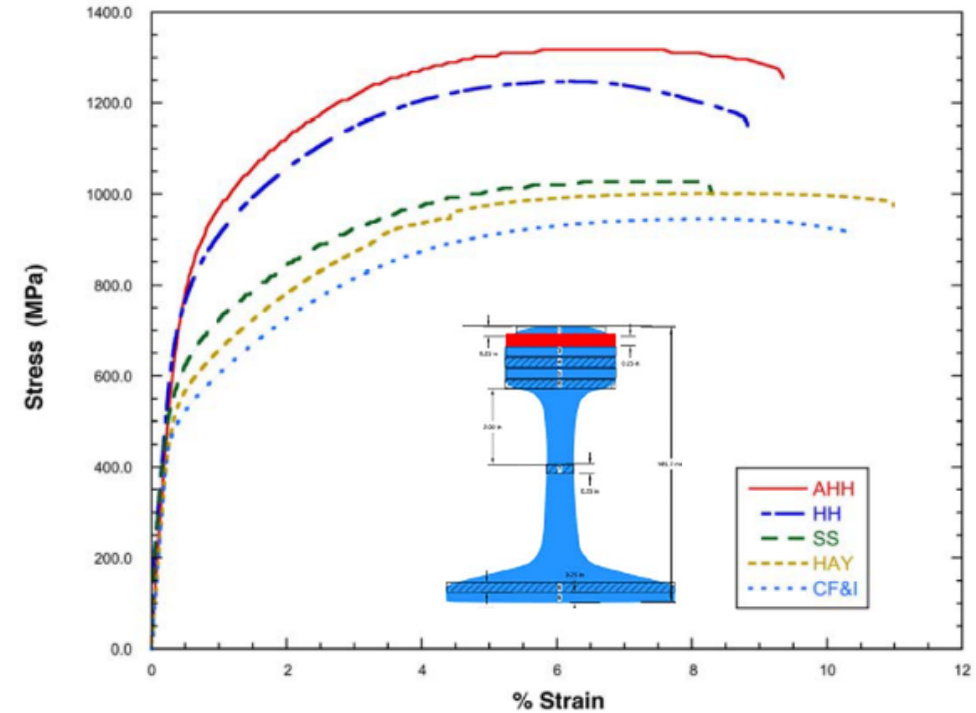
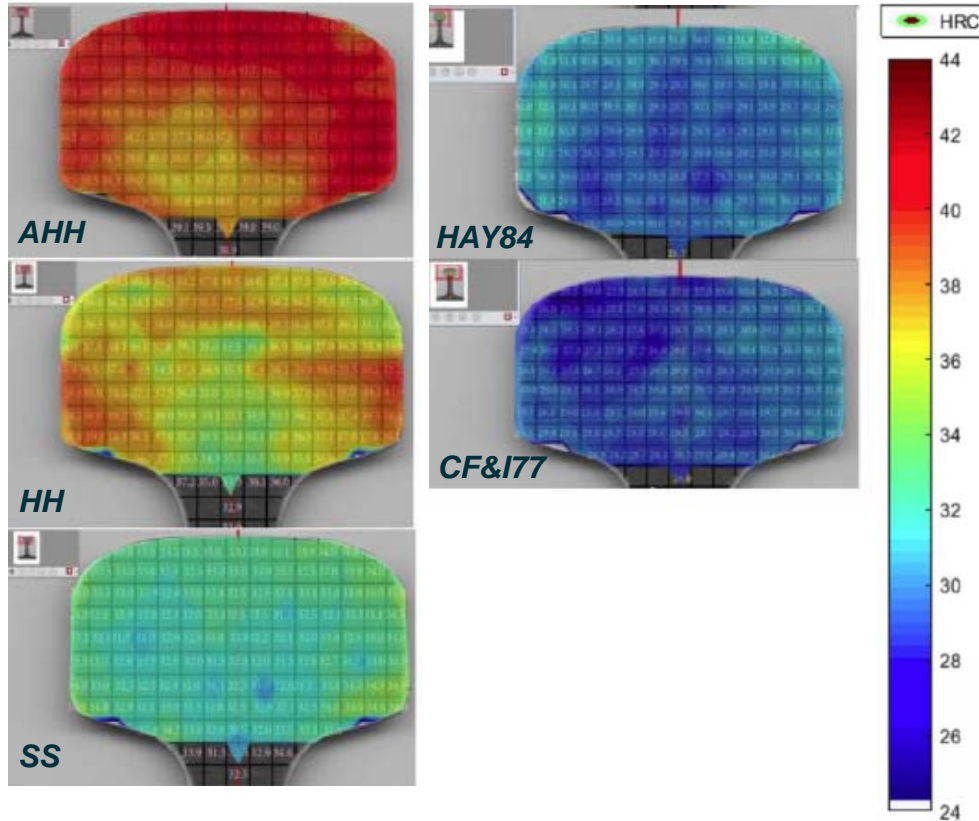


Rail Type	Designation	Manufacturer	Weight	Heat Treatment	Wear
Modern	AHH Advanced Head Hardened	ArcelorMittal	136 RE	Head hardened; fast cooled	New
Modern	HH Head Hardened	ArcelorMittal	136 RE	Head hardened	New
Modern	SS Standard rail	ArcelorMittal	136 RE	Control-cooled	New
Legacy	CF&I77	Colorado Fuel & Iron, 1977	136 RE	Control-cooled	Never used
Legacy	HAY84	Hayange Steel (currently Tata Steel)	136 RE	Vacuum heat treated and degassed	Never used

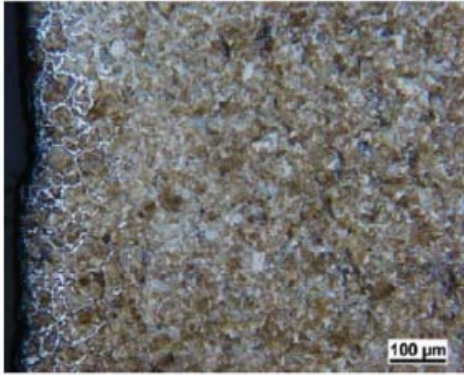


- Current inspection interval based on *LEGACY* rails: 40 MGT
- Modern rails – pearlitic, head-hardened
- *What is the effect of head-hardening on the rails steel properties?*

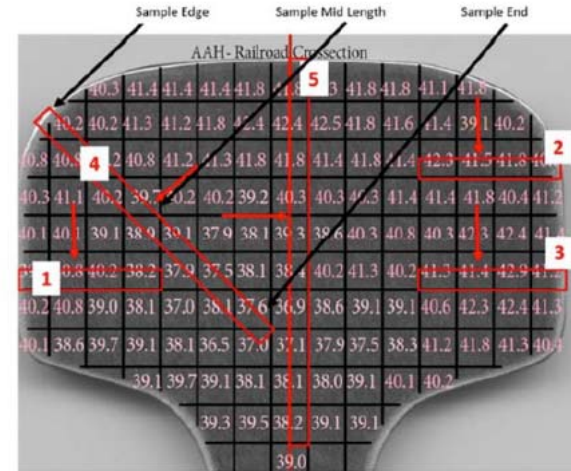
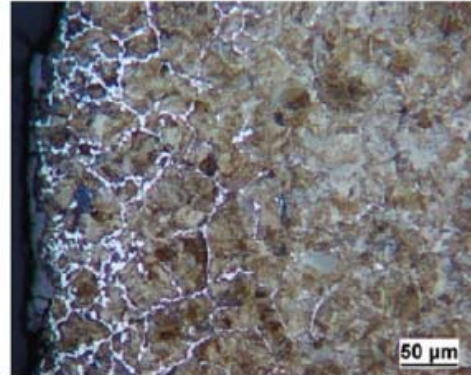
BASIC PROPERTIES



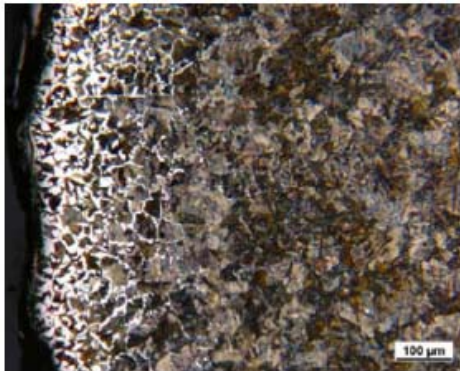
MICROSTRUCTURAL OBSERVATIONS



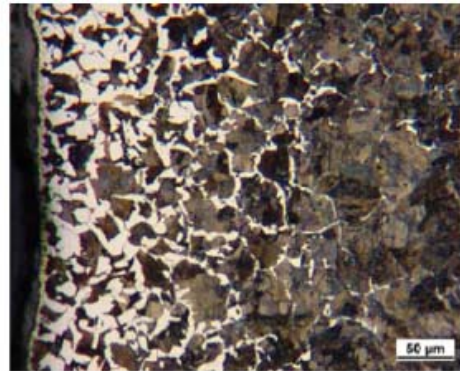
LOM Sample 4 from **AHH** rail at the outer edge of sample



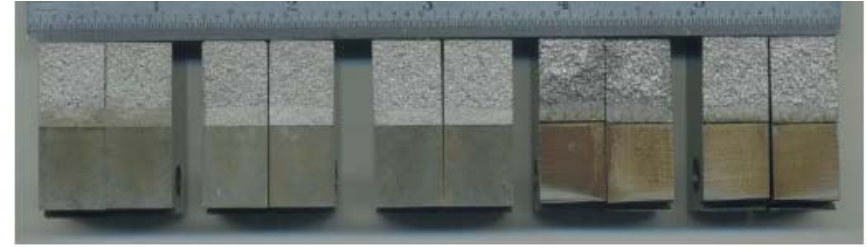
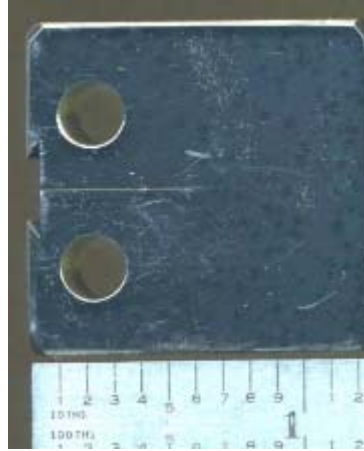
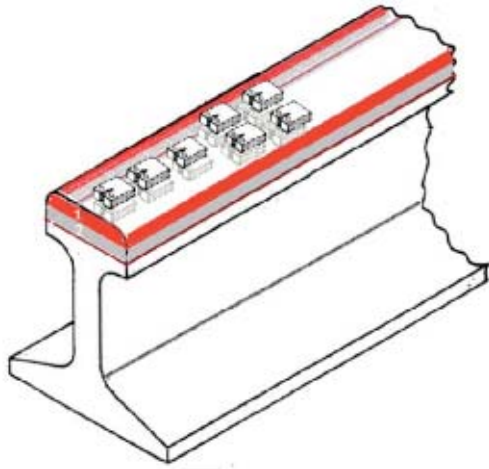
- Uniform pearlitic microstructure, except near the head surface
- Decarburization – carbon depleted zones caused by high temperature processing – not a significant effect



LOM Sample 4 from **HAY84** rail at the outer edge of sample



FRACTURE TOUGHNESS

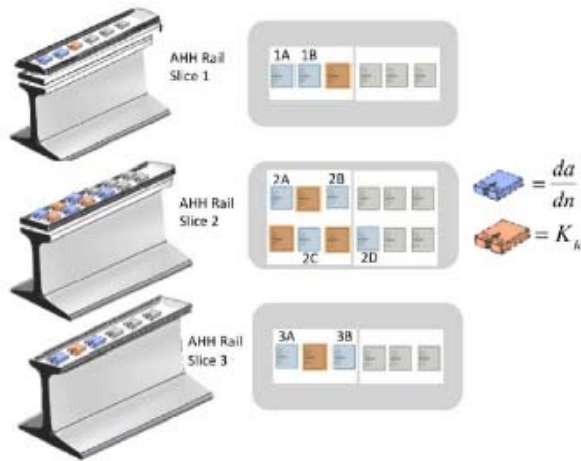


Depth in mm	AHH	HH	SS & HAY84	CF&I77
6.5	40.0	37.1	36.1	44.3
19.5	34.7	36.0	32.8	37.0
32.0	38.8	42.1	36.2	42.6

CT specimens cut from horizontal slices in the rail head

- Relatively small toughness variation across all rails
- Variation within each rail head similar to variation across all rails

CRACK GROWTH RATE



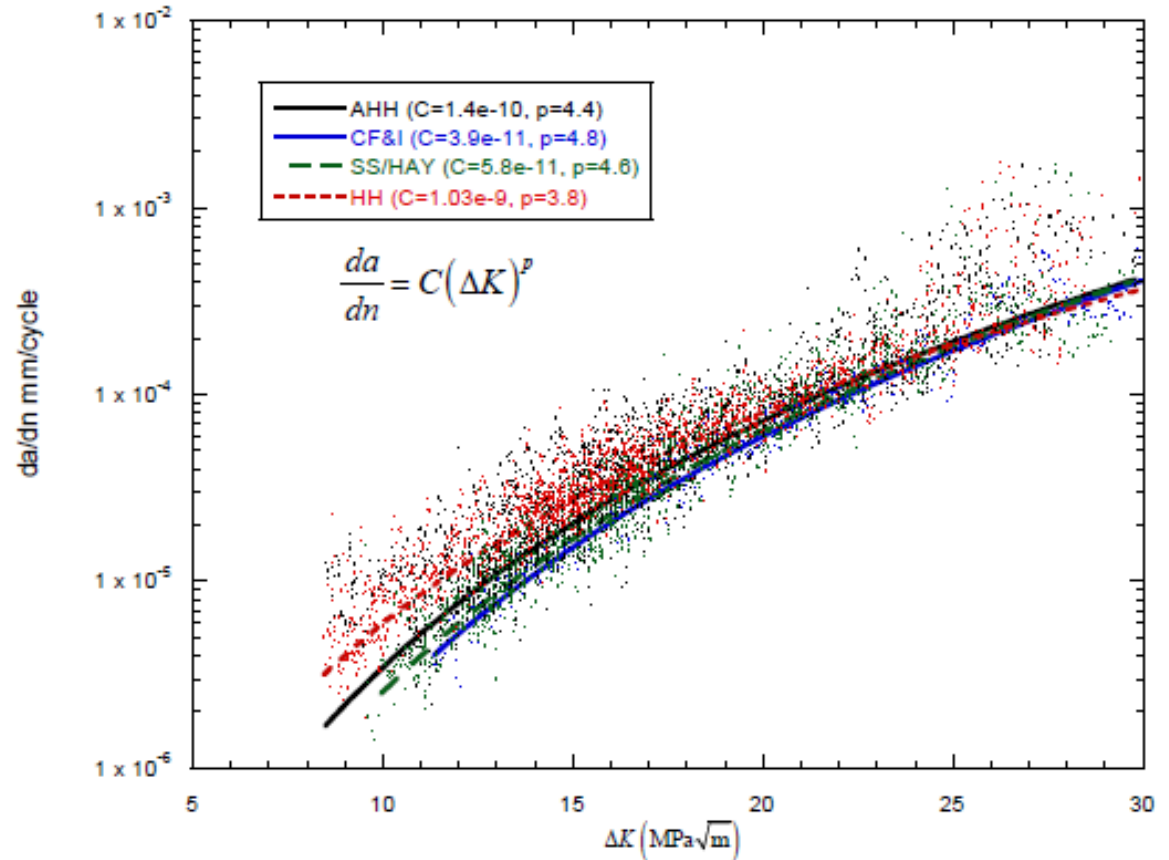
Legacy rails (Jeong et al.)

$$\frac{da}{dN} = C \frac{\Delta K^p}{(1-R)^q} \quad \begin{matrix} C = 10^{-11} \\ p = 4 \\ q = 1.63 \end{matrix}$$

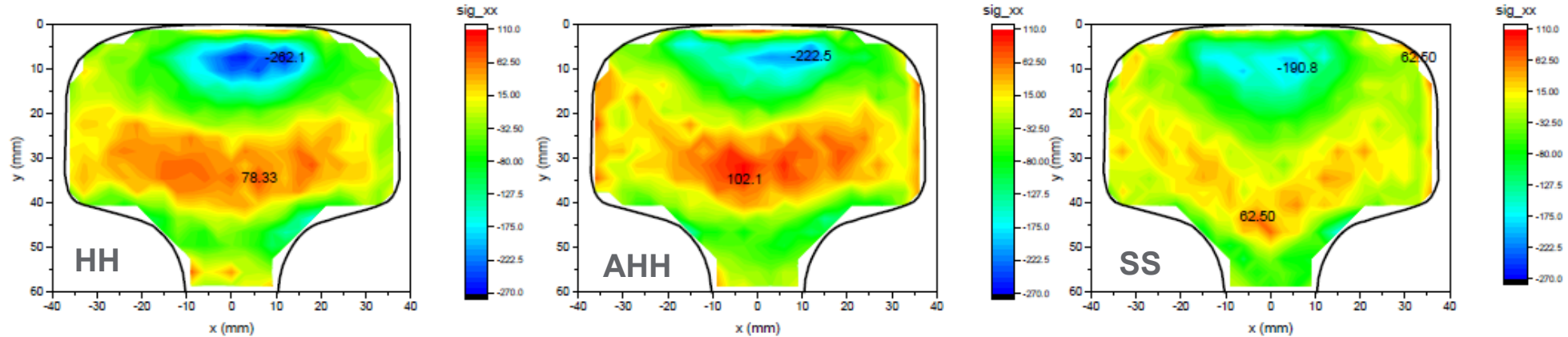
AHH – current study

$$\frac{da}{dn} = C \frac{(\Delta K)^p}{K_c - \Delta K}$$

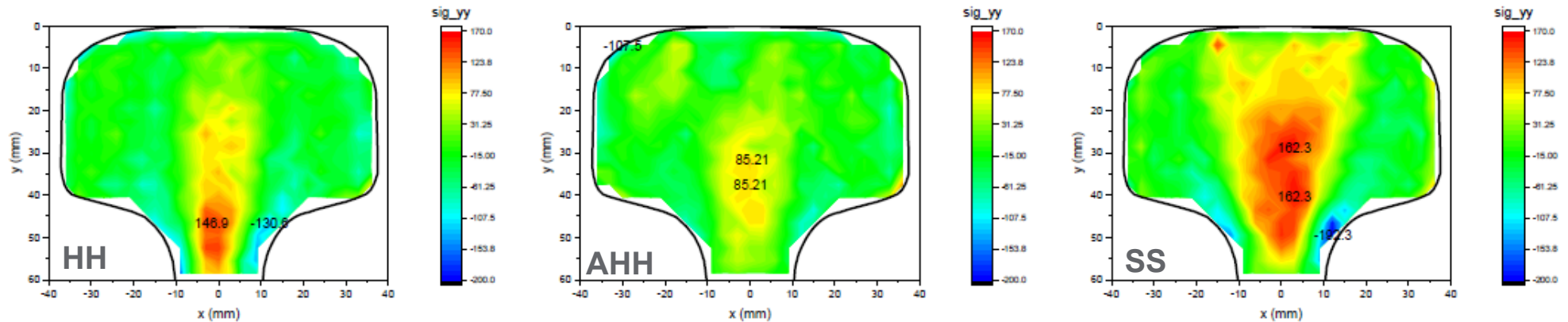
$$C = 3.69 \times 10^{-7}, \quad p = 2.52, \quad K_c = 29.4.$$



RESIDUAL STRESS



Lateral residual stress component σ_{xx} in rails



Transverse residual stress component σ_{yy} in rails

LONGITUDINAL RESIDUAL STRESS (AHH)

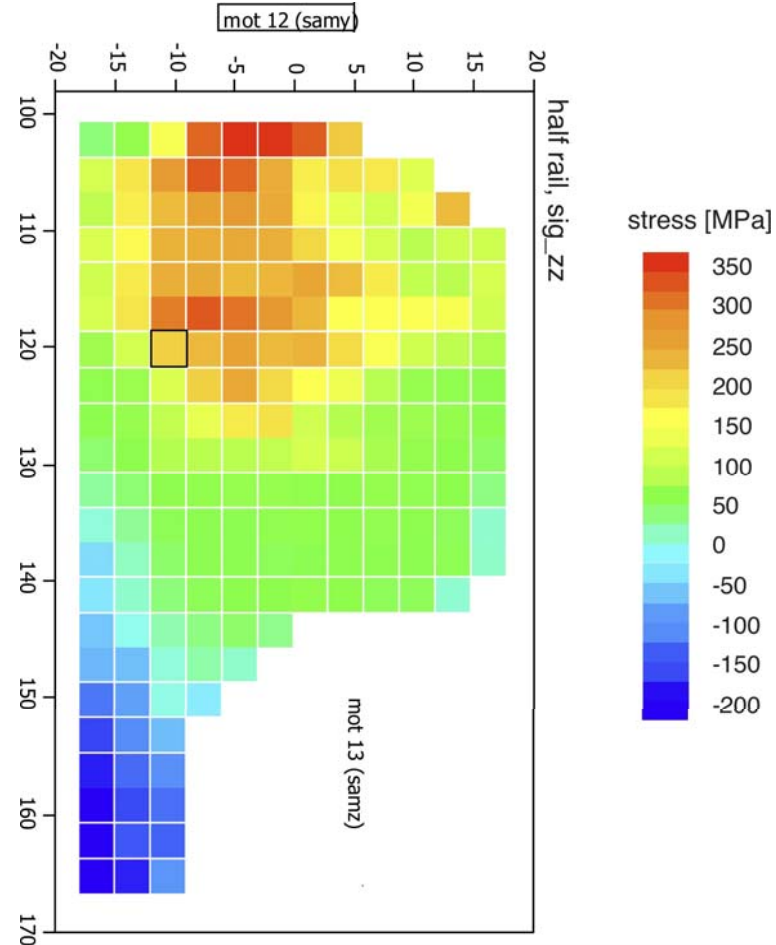


(a)

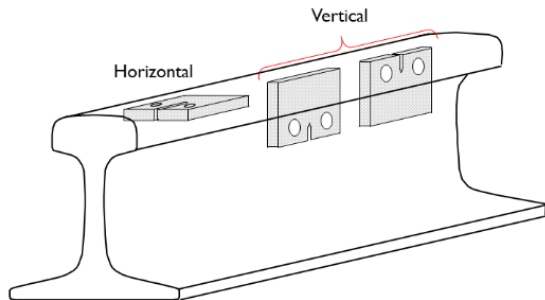


(b)

Specimens for residual stress measurement (a) planar, (b) half rail 300mm long

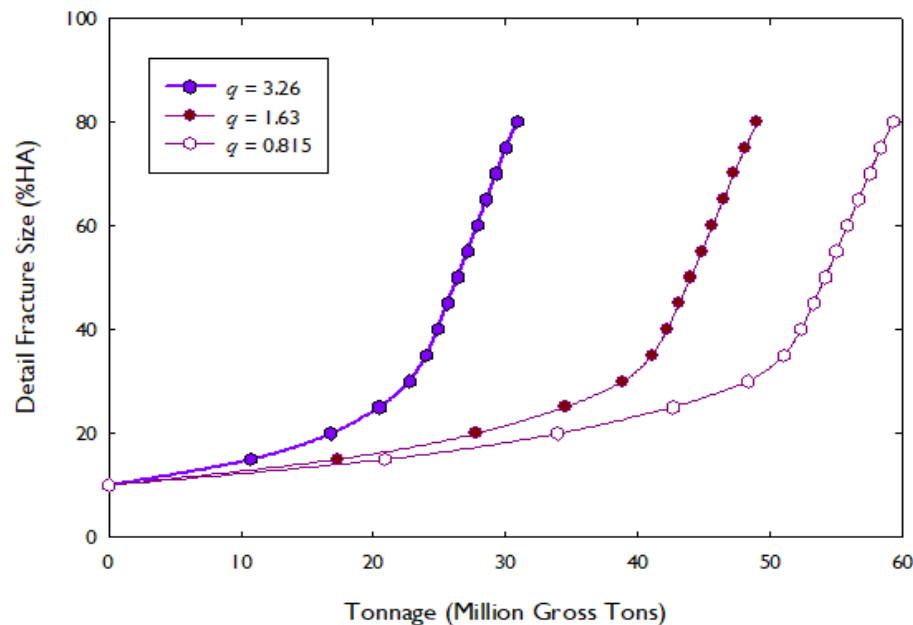
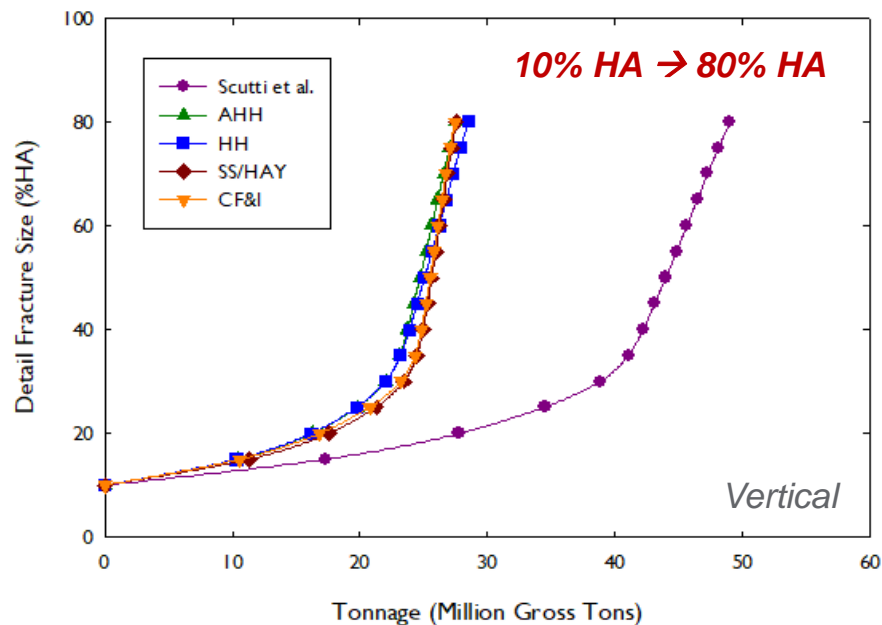


CRACK GROWTH ANALYSIS

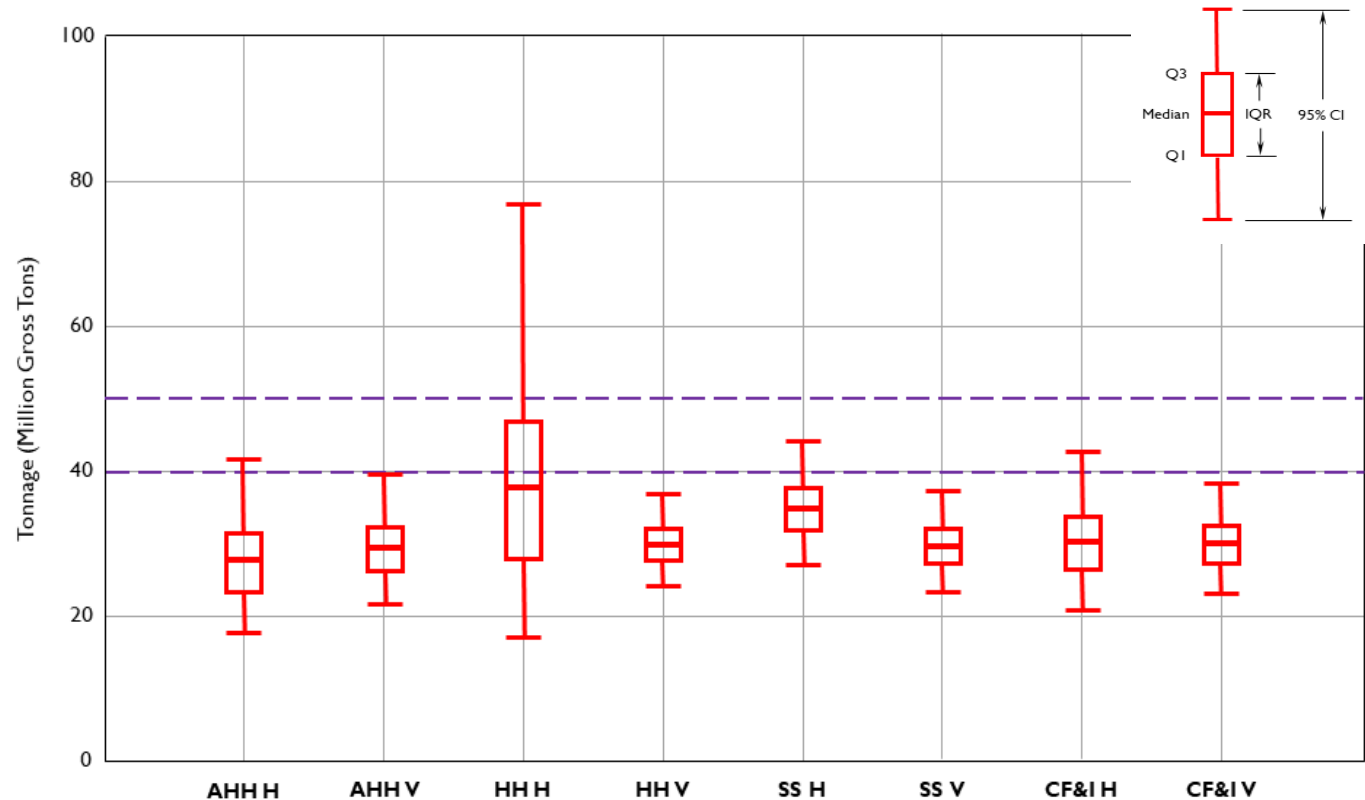
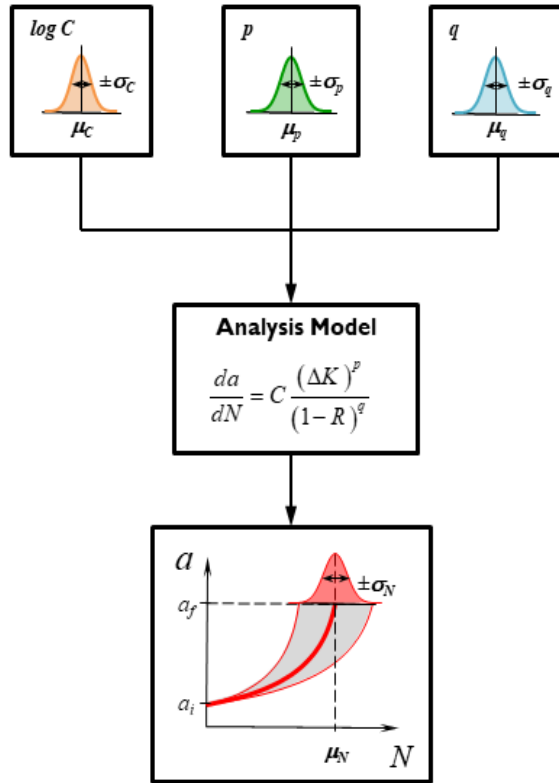


Rail Steel	Slow Crack Growth Life (MGT)
AHH	27.5
HH	28.5
SS/HAY	27.5
CF&I	27.4
Scutti <i>et al.</i>	49.0

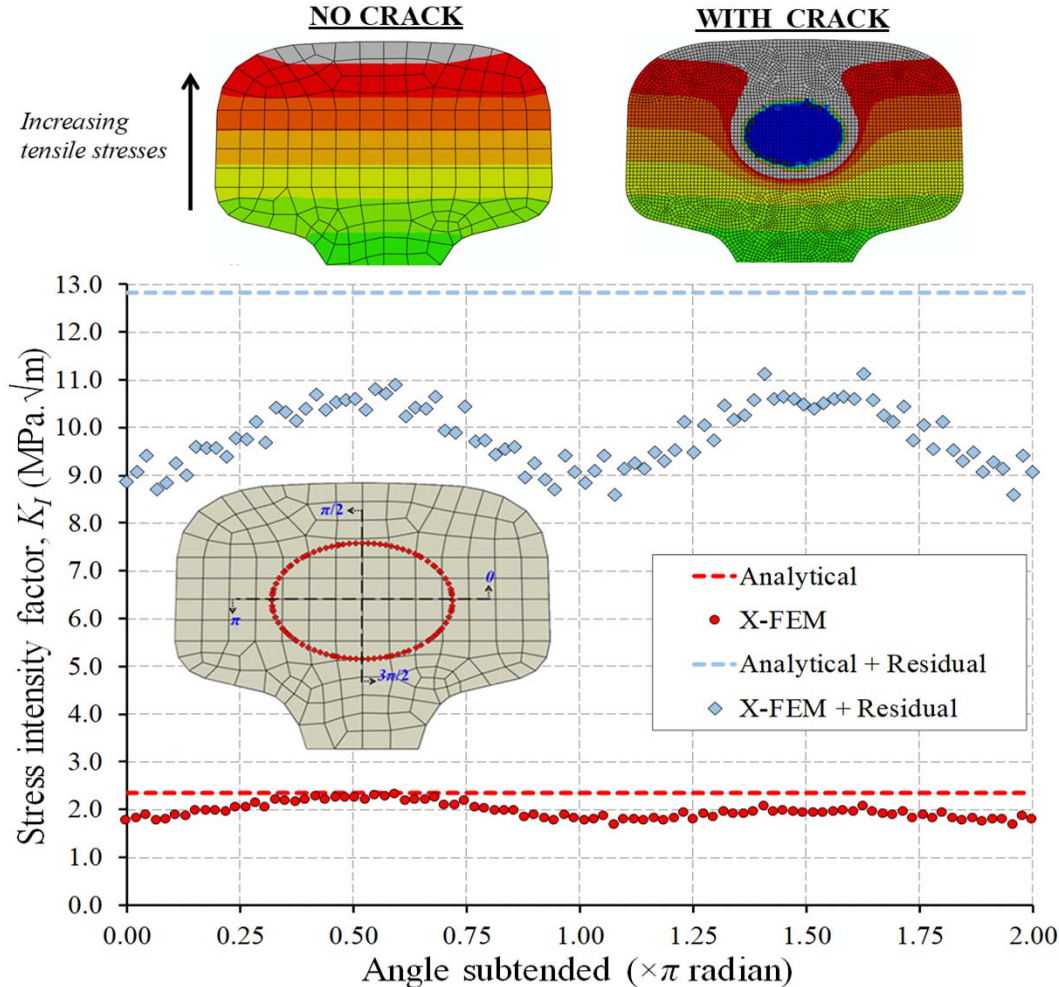
All rails analyzed with the same residual stress



PROBABILISTIC CRACK GROWTH ANALYSIS

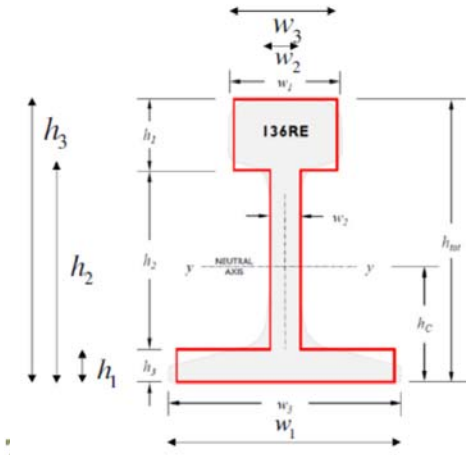
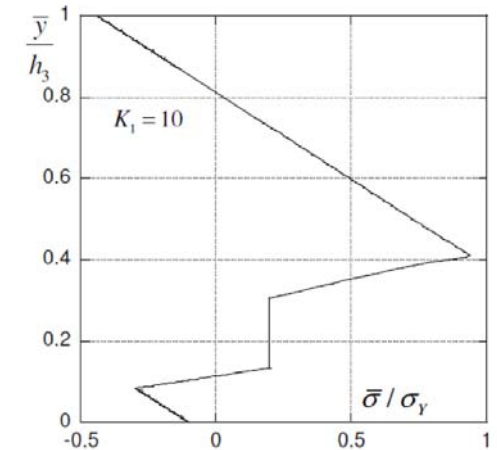
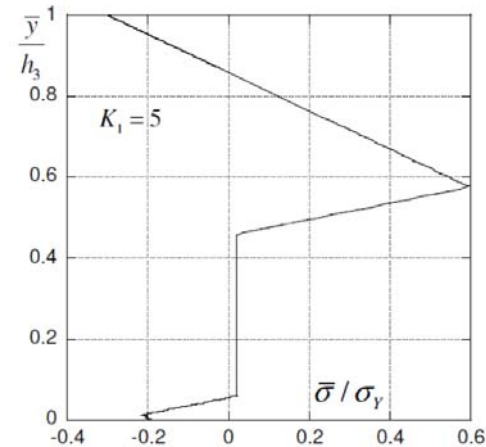
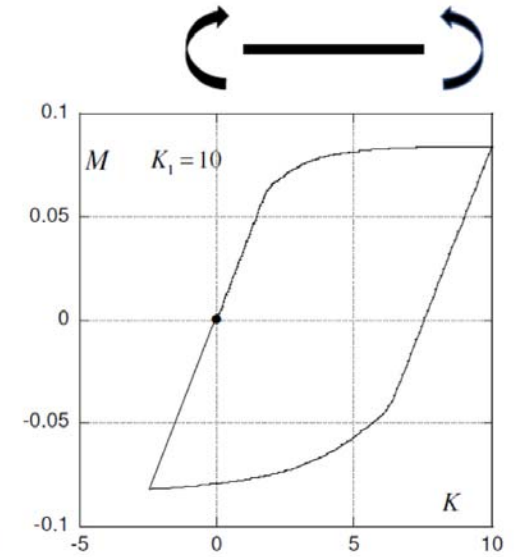
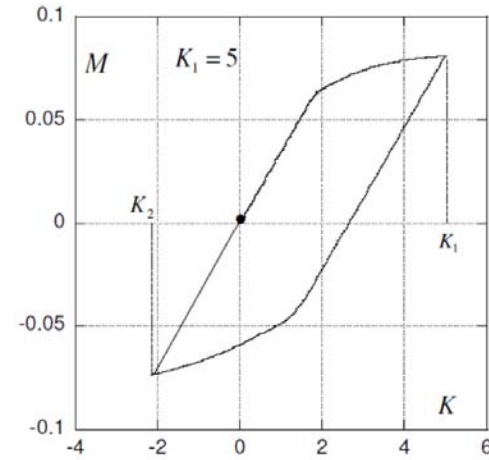


DETAILED ANALYSIS OF THE SIF



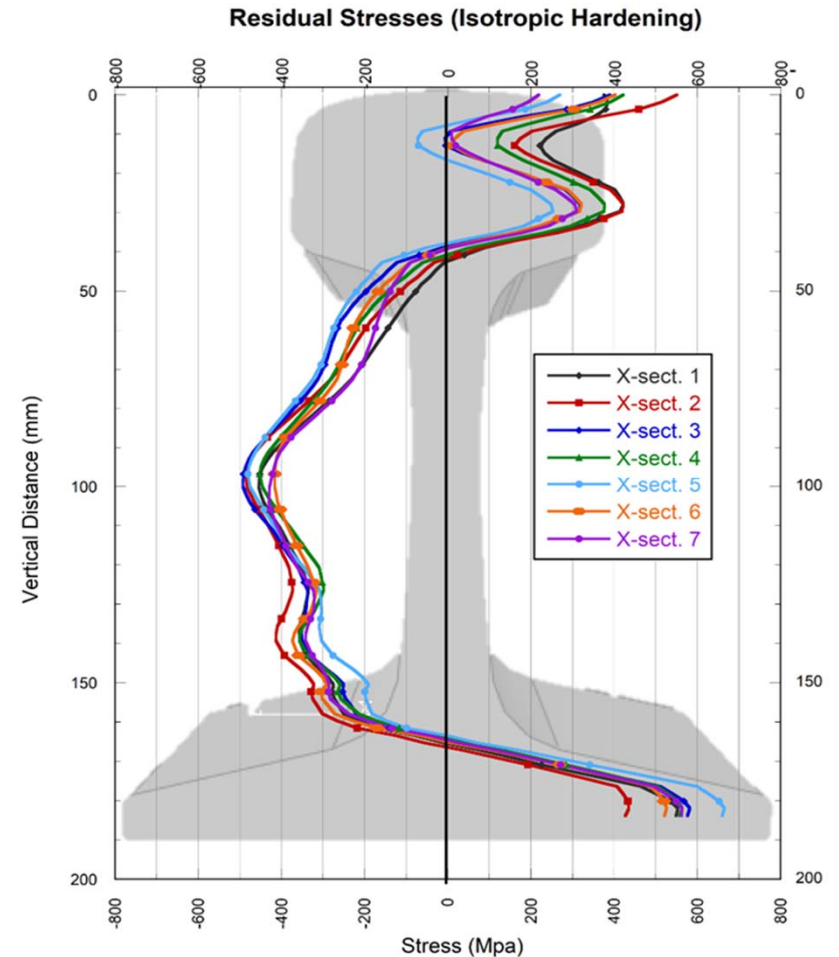
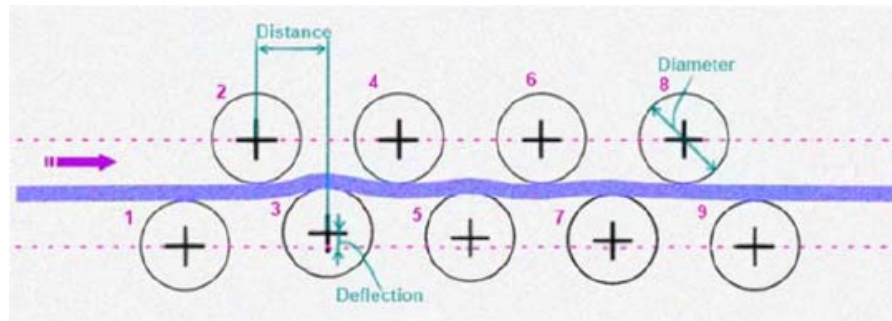
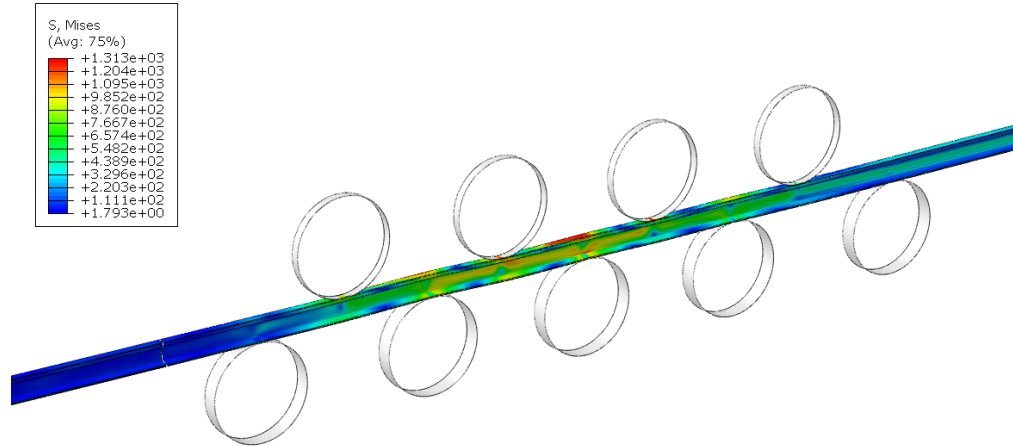
- Analytical model is conservative (as intended)
- Detailed simulation to assess the consequence of initial assumptions
- Improving the accuracy would produce longer life predictions

ROLLER STRAIGHTENING – WIP

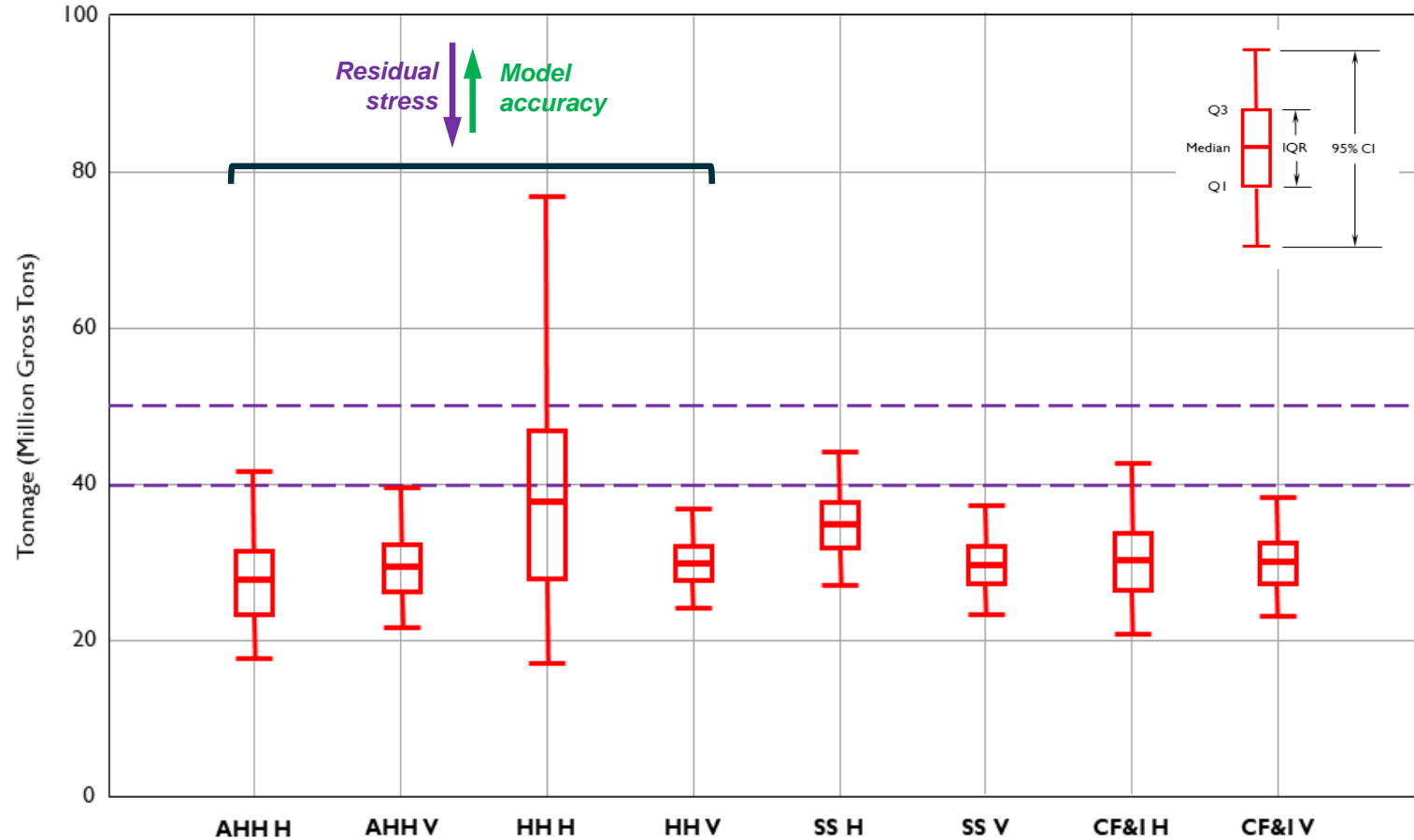


Dimensional
 $\bar{w}_1 = 5.626$ inches, $\bar{h}_3 = 7.312$ inches
 Dimensionless geometry
 $w_1 = 1, w_2 = 0.139, w_3 = 0.474$
 $h_1 = 0.118, h_2 = 0.751, h_3 = 1$
 These refer to the larger font symbols

ROLLER STRAIGHTENING – WIP



RESIDUAL STRESS AND MODEL ACCURACY



SUMMARY – WIP

- Head hardening does not significantly reduce fracture toughness of the rail steel
- Results suggest 30MGT inspection interval likely more appropriate than currently recommended 40MGT (for all rails!)
- Roller straightening is most significant driver of the residual stress: increasing steel strength increases residual stresses
- Theoretically, it is possible to get favourable stress distribution by adjusting the roller straightening process – WIP
- Reducing the level of conservatism in the analytical DF model could offset the effect of faster crack growth due to increased residual stresses – WIP